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Acrylate photopolymerization on heterostructured TiO₂ photocatalysts

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Abstract

Heterostructured TiO_2 samples were prepared by precipitation of Al_2O_3 , Fe_2O_3 hydrate, AgCl and Ag_2O , respectively, onto the surface of a commercial anatase type pigment. The photopolymerization of an ethoxylated trisacrylate was used as a test reaction to check the photocatalytic activity of the heterostructured TiO_2 samples. The TiO_2 sample coated with Al_2O_3 showed no noticeable photocatalytic activity; the untreated TiO_2 was also not a good photocatalyst; in contrast, the TiO_2 samples which had been coated with Fe_2O_3 or AgCl showed enhanced polymerization rate as well as monomer conversion after 120 s illumination. The results of the photopolymerization experiments are explained on the basis of the thermodynamics of the charge transfer between the photoexcited TiO_2 samples and the acrylate. The photoelectric properties of the TiO_2 samples which were characterized using transient Photo-EMF measurements also play an important role in their photocatalytic activity.

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1. Introduction

The photopolymerization of pigmented acrylate formulations is of great importance for coating and printing technologies. In most cases such photopaints consist of a mixture of acrylates, the pigment and molecular photoinitiators such as, benzophenone/amines, triarylphosphinoxides and iodonium salts. UV-illumination of these photopaints leads to fragmentation of the initiating compounds which, in turn, results in the formation of reactive species such as radicals or ions which start the polymerization of the acrylates. Although the use of molecular photoinitiators enables photopaint films to be cured very quickly, the use of molecular photoinitiators has attendant disadvantages:

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fragments of the photoinitiating compounds remain in the polymer film and can migrate;

the photoinitiators and their fragments can impart odour or color changes to the polymer film; and

most photoinitiators are not suitable for application in the fields of foods and medicine.

Clearly, if the pigment itself was able to start photopolymerization by its photocatalytic activity then no molecular photoinitiators should be necessary. Whilst heterogeneous photocatalyzed polymerizations are known in principle [1–7] they are not widely used. Inorganic [2–7] as well as organic pigments [1] have been employed to initiate the photopolymerization of acrylates [1,3], methacrylates [2,4–7] and other vinylics [3]. Although researchers [2,3,5–7] have shown that heterogeneous photocatalyzed polymerization can be carried out in solution, owing to the use of organic solvents such processes are not suited for practical application.

It has been demonstrated [2,3] that nanoparticles of ZnO, TiO₂ and CdS initiate the photopolymerization of the acrylates

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and methacrylates better than particles at the micrometer scale. CdS/HgS and CdS/TiO₂ heterostructures were found [5] to be better photoinitiators than pure CdS.

In order to more fully understand both the findings presented by earlier workers [1–7] and the results contained in this paper, an initial discussion of heterogeneous photocatalytic processes seems sensible. In this context, the general scheme of a heterogeneous photocatalytic reaction is displayed in Fig. 1.

Light absorption by the photocatalyst material leads to an excitation of electrons (e⁻) from the valence band (VB) into the conduction band (CB). As a consequence, positive holes (h⁺) remain in the valence band; electrons and holes can diffuse through the solid followed by trapping or recombination. The generation, diffusion, trapping and recombination of charge carriers due to illumination of the photocatalyst are summarized as photoelectric primary processes. In a second step the charge carriers can be transferred from the catalyst surface to reactants (electron acceptors or donators) such as acrylic monomers.

A photocatalytic reaction is only efficient if electrons and holes are transferred from the catalyst to the reactants by means of suitable electron acceptors and donators, respectively. This charge transfer is possible only if it is thermodynamically possible and if the charge carriers have sufficient lifetime within the surface region. This means that the photoelectric properties of a catalyst material as well as the thermodynamics of charge transfer between the photocatalyst surface and the reactants are important contributors to the efficiency of a heterogeneous photocatalytic reaction. For this reason the

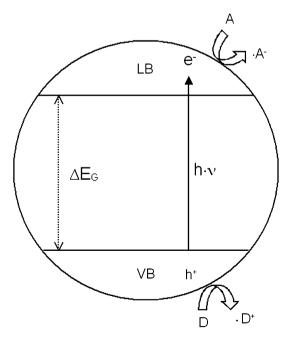


Fig. 1. General scheme of a heterogeneous photocatalytic reaction, VB: valence band, CB: conduction band, ΔE_G : band gap, e⁻: electron, h⁺: positive hole, A: electron acceptor, D: electron donator, $^{\bullet}A^-$: reduced electron acceptor, $^{\bullet}D^+$: oxidized electron donator.

results of experiments on photocatalytic reactions must be discussed in terms of both an energy level scheme and the photoelectric properties of the catalyst materials. However, only a few authors have combined investigations of heterogeneous photocatalytic reactions with measurements of the photoelectric properties of the same charges of material used as catalysts [1,8–11].

In this paper we report the heterogeneous photocatalytic polymerization of an ethoxylated trisacrylate using heterostructured TiO₂. Various inorganic compounds were precipitated (about 1% by mass) onto the surface of a commercial anatase type pigment; for comparison, the untreated commercial anatase type pigment was also investigated. The photoelectric primary processes in the TiO2 samples were characterized using measurements of transient PhotoElectro-Motive Force (Photo-EMF). To perform Photo-EMF measurements the sample is brought into a capacitor and is illuminated by a laser flash through a transparent electrode. The photovoltage is measured as a function of time without any external electric field. As the charge carrier concentration gradient due to the gradient of light absorption and heterojunctions within the sample are the only driving forces for the Photo-EMF generation, this method is well suited to the investigation of influences of the structure of the materials on their photoelectric properties [12].

The amount of the maximum Photo-EMF, $U_{\rm max}$ is a measure of the efficiency of charge separation upon illumination. The Photo-EMF decreases due to recombination or chemical reaction of the charge carriers. The Photo-EMF decay is expressed by a biexponential rate law (Eq. (1)):

$$U(t) = U_1^0 \exp(-k_1 t) + U_2^0 \exp(-k_2 t)$$
 (1)

By definition, the process with parameters U_1^0 and k_1 is always the faster decay process so that $k_1 > k_2$. In previous works, the faster decay process was assigned to a Photo-EMF in the subsurface region [13,14]; the Photo-EMF method is described in more detail elsewhere [11,13].

2. Experimental

2.1. Materials

The base material was an uncoated anatase type TiO_2 pigment grade K1001 made by Kronos; TiO_2 coated with Al_2O_3 (grade K1014) was also supplied by Kronos. $FeCl_3 \cdot 6H_2O$, AgNO₃, NaOH, NaCl, 1,2-dichlorethane and dichloromaleic acid anhydride were purchased from Merck. The pigments and chemicals mentioned above were used as received. The acrylate used (grade SR415) was purchased from Cray Valley; prior to use, inhibitors were removed by filtration over an Al_2O_3 column.

2.2. Coating of TiO_2 with Fe_2O_3 hydrate

 $101 \text{ mg FeCl}_3 \cdot 6H_2O$ was dissolved in 500 ml distilled water; in this solution 2 g of the TiO_2 pigment was dispersed.

The mixture was stirred for 24 h at room temperature and the pigment was then separated by centrifugation (10 min, 5000 rpm) and washed three times with distilled water.

The ${\rm TiO_2}$ loaded with ${\rm Fe^{3+}}$ ions was dispersed in 200 ml of 0.1 N NaOH. After immersion into the NaOH the pigment becomes immediately brownish showing the presence of ${\rm Fe_2O_3}$ hydrate. After 1 h of stirring at room temperature, the pigment was separated by centrifugation (10 min, 5000 rpm), washed with distilled water and dried in a desiccator. The coated ${\rm TiO_2}$ (1 g) was heated to 220 °C/24 h to transform the ${\rm Fe_2O_3}$ hydrate into ${\rm Fe_2O_3}$.

2.3. Coating of TiO₂ with AgCl and Ag₂O, respectively

The coating of TiO₂ with AgCl as well as Ag₂O was carried out in the absence of daylight. For coating with AgCl, 12 mg of AgNO₃ and for coating with Ag₂O 15 mg of AgNO₃ were dissolved in 100 ml distilled water. In each case, 1 g of the TiO₂ pigment was dispersed in the ensuing solution. The mixture was stirred for 24 h at room temperature and the pigment was then separated by centrifugation (10 min, 5000 rpm) and washed with distilled water. The Ag⁺ ion loaded TiO₂ samples were immersed in 100 ml of 0.1 N NaCl and 0.1 N NaOH, respectively, and stirred for 1 h at room temperature, after which, they were separated by centrifugation (10 min, 5000 rpm), washed with distilled water and dried at 110 °C for 3 h.

2.4. Determination of the amounts of coating materials deposited on the TiO_2 surface

To determine the amounts of Fe³⁺ as well as Ag⁺ ions deposited on the TiO₂ surface, the Fe³⁺ as well as the Ag⁺ content of the solution was measured after the adsorption experiment. The Fe³⁺ content was determined photometrical using a Shimadzu UV-3101 PC, employing complex forming reaction with SCN⁻. The Ag⁺ content was determined by means of anodic stripping voltammetry using a glassy carbon

patterns were recorded on a Phillips X'Pert Pro MPD X-ray diffractor in the range of 2Θ between 10° and 70° using Cu K α_1 irradiation. A combination of scanning electron microscopy (SEM) and EDX (Oxford Instruments) was used to investigate the morphology of the TiO_2 and to detect the coating materials.

2.6. Photo-EMF measurements

Photo-EMF measurements were undertaken using pigment polymer dispersion layers, employing polyvinyl butyral as polymeric binder. The layers were prepared by a solution casting process using 1,2-dichlorethane as solvent. The dispersion layers used contained about 25% pigment by mass, were $60-80~\mu m$ thick and displayed total absorption in the UV range.

From the layer, pieces measuring 10 mm in diameter were cut and brought into the Photo-EMF device. The sample was illuminated by a single flash of a nitrogen laser PNL 100 made by Lasertechnik Berlin GmbH (wavelength: 337 nm, pulse duration: 300 ps, power: 100 kW). The light intensity at the sample's place was about 3×10^{13} quanta per flash; the temperature of the sample and the amplifier was 25 °C.

All Photo-EMF signals and parameters presented herein were the mean values of three measurements; for each measurement a new piece of the layer was used. The Photo-EMF device is constructed like a capacitor with one transparent electrode. The Photo-EMF is measured without any external electric field (i.e. contactless) by insulating foils between the sample and the electrodes; see Ref. [13] for further details of the Photo-EMF device and the measurement procedure used.

2.7. Photopolymerization

Polymerization of the trisacrylate 1 was used to investigate the photocatalytic activity of the TiO_2 samples. As a promoting agent 6.5 mol% of the electron acceptor dichloromaleic acid anhydride 2 was added to the acrylate (see Section 3.4).

electrode as working electrode, Ag/AgCl as reference electrode and Pt-wire as counter electrode.

2.5. Structural analyses

Powder X-ray diffraction (XRD) was used to detect the crystal type of the TiO₂ samples. The X-ray diffraction

The sample contained 5.5% by mass of the TiO_2 investigated; as a comparison, polymerization of the mixture was attempted without TiO_2 . For the polymerization experiments, films of the pigment monomer mixture of layer thickness $20 \, \mu m$ were prepared using a knifecoater. The layers were illuminated by white light from a Mercury high pressure lamp (power $100 \, W$); to avoid direct excitation of the acrylate,

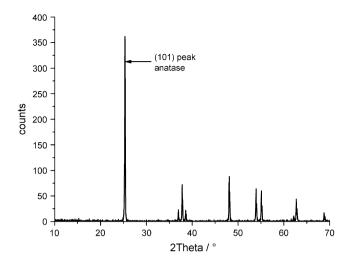


Fig. 2. X-ray diffraction pattern of a powder sample of untreated TiO₂.

radiation with wavelengths shorter than 350 nm was removed using a cut-off filter. The light intensity at the sample was 90 mW cm^{-2} .

The relative concentration of double bonds was measured by real time infrared spectroscopy using a Biorad FTS6000 spectrometer; the intensity of the absorption band at 810 cm⁻¹ (C–H wagging vibration of an unsaturated carbon atom) was monitored as a function of illumination time. To exclude oxygen from air the sample chamber was purged with nitrogen 3 min before the measurement begun as well as during the measurement procedure. All results of the polymerization experiments presented here are the mean values of three attempts.

3. Results and discussion

3.1. Amount of the coating materials deposited on the TiO_2 surface

The analyses described in Section 2.4 provided the following results. After the adsorption process no Fe³⁺ ions were detected in solution from which it can be concluded that the

full amount of Fe^{3+} ions had been adsorbed onto the TiO_2 surface, which corresponds to 1% by mass of Fe_2O_3 .

In the wash water no Ag^+ ions were detected and the supernatant contained 4.4% of the original Ag^+ used in the AgCl coating and 23.8% of the original Ag^+ used in the Ag₂O coating. These values correspond to an AgCl quantity of 0.97% by mass and an Ag₂O quantity of 0.78% by mass on the TiO₂, respectively. The Al₂O₃ content of the commercial Al₂O₃ coated TiO₂ amounts to 1.4% by mass according to the supplier.

3.2. Structure of the TiO₂ samples

The X-ray diffraction pattern of the neat TiO_2 showed a diffraction peak at $2\Theta = 25.33^{\circ}$ which is the (101) diffraction of the anatase type (Fig. 2); because at $2\Theta = 27.42^{\circ}$ there was no diffraction peak, there was no rutile form of the pigment present in the sample.

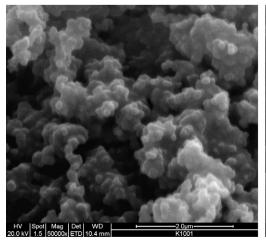
The X-ray diffraction patterns of the coated samples agree completely with that of the untreated sample which shows that the coatings cannot be detected using XRD. One reason for this is the amorphous nature of the coatings in the cases of the Al_2O_3 and Fe_2O_3 hydrates. Although the Fe_2O_3 , AgCl and Ag_2O should be crystalline, the amounts of the coating materials in the samples were below the detection level of XRD.

Fig. 3a shows an SEM micrograph of the untreated TiO_2 which reveals the spherical primary particles having sizes of about 200 nm formed large agglomerates. The presence of the coating materials did not change the morphology of the TiO_2 , as exemplified by the SEM micrograph of the TiO_2 sample coated with Fe_2O_3 shown in Fig. 3b.

The presence of the coating materials Al, Fe or Ag on the TiO₂ samples was proven qualitatively by means of EDX. The EDX analysis of the neat TiO₂ revealed that this sample consists only of Ti and O; impurities could not be detected.

3.3. Photoelectric properties

Fig. 4a shows the Photo-EMF signals of the TiO₂ samples recorded in the time range of up to 100 ms after the laser flash.



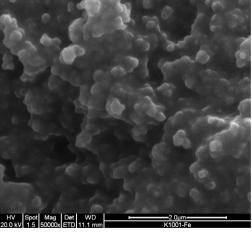
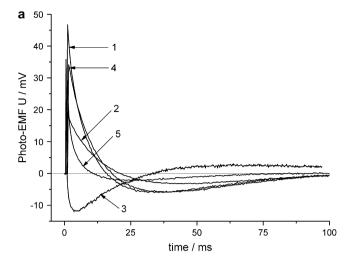


Fig. 3. SEM micrograph of a powder sample of (a) untreated TiO2 and (b) TiO2 coated with 1% by mass Fe2O3.



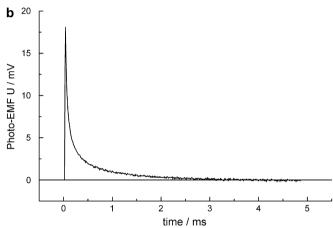


Fig. 4. (a) Photo-EMF signals of coated TiO_2 samples in comparison to untreated TiO_2 , excitation wavelength: 337 nm, laser flash intensity at the samples place: 3×10^{13} quanta per flash, curve 1: neat TiO_2 , curve 2: 1.4% by mass Al_2O_3 on TiO_2 , curve 3: 1.0% by mass Fe_2O_3 on TiO_2 , curve 4: 0.97% by mass Ag_2O on TiO_2 , curve 5: 0.78% by mass Ag_2O on TiO_2 . (b) Photo-EMF signal of a TiO_2 sample coated with 1.0% by mass Fe_2O_3 hydrate.

As the time range shown in Fig. 4a is not suited to visualize the Photo-EMF decay of the TiO_2 sample coated with 1.0% by mass Fe_2O_3 hydrate, the signal is shown in a separate figure (Fig. 4b). The parameters of the signals shown in Fig. 4a and b are summarized in Table 1.

The results shown in Fig. 4a, b and Table 1 can be summarized as follows. All of the TiO_2 samples show Photo-EMF signals starting with a positive sign which means that all of the samples behaved as n-type photoconductors which is typical for TiO_2 . Whilst the coating materials used in this work were not able to change the type of photoconduction in TiO_2 , the coating materials, nevertheless, changed the photoelectric properties of the TiO_2 although the amounts of those materials are low. Generally, the maximum values $U_{\rm max}$ for the Photo-EMF of the coated samples were lower than that of the untreated sample. Deposition of Al_2O_3 caused the strongest decrease of $U_{\rm max}$; this decrease could signify a reduction in the efficiency of charge separation due to the coating. Another reason for a decrease in $U_{\rm max}$ could be charge transfer between the TiO_2 and the coating material.

Although from the values of $U_{\rm max}$, it can only be concluded that the coatings should change the photocatalytic activity of TiO₂, it is difficult to predict if the photocatalytic activity was increased or decreased by coating. The coating of TiO₂ strongly influenced the Photo-EMF decay kinetics: deposition of Fe₂O₃ hydrate increased k_1 by two orders of magnitude whereas in contrast, coating with Al₂O₃, AgCl or Ag₂O led to a decrease in k_1 . The finding that in general, the coating of TiO₂ altered the value of k_1 more than the value of k_2 , confirms our previous result that k_1 may be assigned to a Photo-EMF in the subsurface region.

The TiO_2 sample coated with 1.0% by mass Fe_2O_3 showed a complex Photo-EMF signal comprising two crossing points with zero potential; such behaviour may be a hint at a heterojunction in the sample.

3.4. Photopolymerization

Fig. 5 shows the relative concentration of double bonds as a function of illumination time for samples consisting of the acrylate 1, the electron acceptor 2 and the TiO_2 samples mentioned; the kinetics of the photopolymerization of a blank sample is shown too.

If the pigment is photocatalytic active the pigmented sample must polymerize faster than the reference without pigment.

The maximum polymerization rate $r_{\rm P}^{\rm max}$ was determined by differentiating the curves shown in Fig. 5. Moreover, the double bond conversion after an illumination time of 120 s $C_{120~\rm s}$ was determined from the experimental curves. The results are summarized in Table 2.

The sample pigmented with neat TiO_2 polymerized faster than the pigment free reference but the difference was not very large. This shows that the neat TiO_2 has only small photocatalytic activity; this behaviour was expected because this pigment is not optimised for application in heterogeneous photocatalysis.

Coating TiO_2 with Al_2O_3 lowered the photopolymerization rate in comparison to neat TiO_2 , corresponding to the results discussed previously [15]. Coating with Al_2O_3 suppressed the photocatalytic activity of TiO_2 and, therefore, TiO_2 coated with Al_2O_3 can be used for applications where the photocatalytic activity of TiO_2 would be detrimental.

In the case of the sample containing TiO_2 coated with Fe_2O_3 hydrate the monomer conversion was rather low in comparison to the reference. This sample showed nearly no photocatalytic activity but acted as a light filter which suppressed homogeneous polymerization. This poor photocatalytic behaviour can be explained by the extreme short charge carrier lifetimes in this sample, as described in Section 3.3. So the charge carriers recombine before they can be transferred to the monomer.

If the Fe₂O₃ hydrate on top of the TiO₂ is converted into Fe₂O₃ by a heating process, a remarkable increase in polymerization rate $r_{\rm P}^{\rm max}$ and monomer conversion $C_{120~\rm s}$ is observed, showing that a photocatalytic inactive material can be changed into an active one in a simple way. The TiO₂ coated with AgCl shows the best photocatalytic activity: In comparison to all

Table 1 Maximum values $U_{\rm max}$ and kinetic parameters of the Photo-EMF curves shown in Fig. 4a and b

Curve no. in Fig. 4a	Sample	U _{max} (mV)	$U_1^0 \text{ (mV)}$	$U_2^0 \text{ (mV)}$	$k_1 (s^{-1})$	$k_2 (s^{-1})$
1	Neat TiO ₂	46.7 ± 3.3	63 ± 7	-16 ± 2	117 ± 1.2	26.2 ± 0.3
2	$TiO_2 + 1.4 \text{ wt}\% \text{ Al}_2O_3$	19.2 ± 1.4	255 ± 28	-253 ± 32	44.9 ± 0.5	44.5 ± 0.5
Curve in Fig. 4b	$TiO_2 + 1$ wt% Fe_2O_3 hydrate	28.5 ± 2.0	24 ± 3	4 ± 0.5	$(2.1 \pm 0.1) \times 10^4$	$(1.6 \pm 0.1) \times 10^3$
3	$TiO_2 + 1$ wt% $Fe_2O_3^a$	36.0 ± 2.5	38 ± 4	-2 ± 0.2	$(2.7 \pm 0.1) \times 10^3$	65.7 ± 0.7
4	$TiO_2 + 0.97$ wt% AgCl	34.1 ± 1.9	5052 ± 1662	-5018 ± 1661	53.5 ± 0.3	53.2 ± 0.4
5	$TiO_2 + 0.78 \text{ wt}\% \text{ Ag}_2O$	30.3 ± 1.8	47 ± 5	-16 ± 4	96.3 ± 6.6	31.6 ± 1.6

^a Photo-EMF signal contains two crossing points with zero potential. For that reason it cannot be described completely by Eq. (1). The parameters mentioned in Table 1 describe the signal from its maximum value up to the second crossing point with zero potential.

other samples investigated in this work, both the polymerization rate and the monomer conversion were highest.

Thus, the results presented in this work show that the photocatalytic activity of TiO_2 can be enhanced or suppressed by coating with suitable materials. To understand the results of the photopolymerization experiments the thermodynamics of charge transfer between the photoexcited TiO_2 and the acrylate must be taken into account. In Fig. 6 the energy levels of the valence and conduction band edges as well as the band gap energies ΔE_G are shown for TiO_2 and the coating materials used in this work [16–21]. Moreover, Fig. 6 shows the energetic positions of the highest occupied molecular orbital (HOMO) and the lowest unoccupied molecular orbital (LUMO) of the acrylate.

The shortest light wavelength used was 350 nm, meaning that the highest available photon energy was 3.5 eV. Whilst this energy was not sufficient to transfer electrons from the HOMO to the LUMO of the acrylate, it was sufficiently high for interband excitation to occur in the TiO₂. The upper edge of the valence band of TiO2 lies below the HOMO of the acrylate; this means that holes from the valence band of the photoexcited TiO₂ can be transferred to the HOMO of the acrylate. This process leads to oxidation of the acrylate and, in this way, radical cations are formed which can initiate polymerization. However, TiO₂ is not able to transfer electrons from its conduction band into the LUMO of the acrylate because the LUMO of the acrylate is located at a higher energy level than the conduction band of the TiO₂. Consequently, an electron acceptor such as dichloromaleic acid anhydride must be added to trap the electrons in the TiO₂. Indeed, without the electron acceptor no remarkable photopolymerization of the acrylate is observed in the presence of TiO₂ because the photocatalytic process is stopped by the negative charging of the TiO_2 .

The light used is not able to excite electrons from the valence band into the conduction band of Al_2O_3 . Moreover, an electron or hole transfer between Al_2O_3 and the photoexcited TiO_2 is forbidden thermodynamically, because the valence band of Al_2O_3 lies below that of the TiO_2 and the conduction band of Al_2O_3 lies above that of TiO_2 . Hence, the Al_2O_3 does not contain any free charge carrier which could be transferred to the acrylate. However, the efficiency of direct charge transfer between the TiO_2 and the acrylate is reduced because the

 Al_2O_3 partially covers the active TiO_2 surface and so no charge transfer can occur at the covered places between the TiO_2 and the reactants.

In the case of TiO₂ coated with Fe₂O₃, AgCl or Ag₂O, the incident light causes interband excitation of electrons in the TiO₂ as well as in the coating. Here the different positions of the energy bands of the coatings govern the photocatalytic activity of the coated samples. Coating with Fe₂O₃ increases the photocatalytic activity of TiO₂ although the upper edge of valence band of Fe₂O₃ is located above the HOMO of the acrylate. However, in this case, it must be considered that in Fe₂O₃ the incident light is able to promote electrons into the conduction band which are located 1.2 eV below the upper edge of the valence band, which means that electrons have an energy level of about -8.2 eV. Consequently in the photoexcited Fe₂O₃ coating, holes in the valence band having energy levels above -8.2 eV are available which can be transferred to the acrylate to form initiating radical cations. Thus, the TiO₂ as well as the coating material are able to oxidize the acrylate, resulting in enhanced photocatalytic activity.

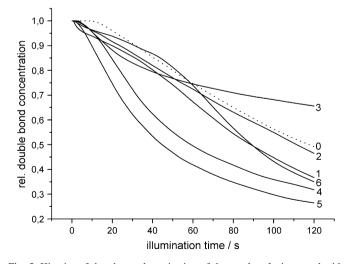


Fig. 5. Kinetics of the photopolymerization of the acrylate 1 pigmented with 5.5% by mass of the TiO $_2$ samples mentioned (solid lines): curve 1: neat TiO $_2$, curve 2: 1.4% by mass Al $_2O_3$ on TiO $_2$, curve 3: 1.0% by mass Fe $_2O_3$ hydrate on TiO $_2$, curve 4: 1.0% by mass Fe $_2O_3$ on TiO $_2$, curve 5: 0.97% by mass AgCl on TiO $_2$, curve 6: 0.78% by mass Ag $_2O$ on TiO $_2$. The dotted line shows the polymerization kinetics of the acrylate without TiO $_2$ but in the presence of the electron acceptor 2 (curve 0).

Table 2 Maximum rate of the photopolymerization $r_p^{\rm max}$ and double bond conversion after 120 s illumination $C_{120~\rm s}$ of the acrylate samples mentioned in Fig. 5

Curve no. in Fig. 5	Sample	$r_{\text{max}}^{\text{P}}/[\text{M}]_0 \text{ (s}^{-1})$	$C_{120 \text{ s}} (\%)$
0	Without TiO ₂	$(5.5 \pm 0.3) \times 10^{-3}$	50.8 ± 2.5
1	Neat TiO ₂	$(6.8 \pm 0.3) \times 10^{-3}$	63.3 ± 3.2
2	$TiO_2 + 1.4 \text{ wt}\% \text{ Al}_2O_3$	$(5.0 \pm 0.3) \times 10^{-3}$	53.5 ± 2.7
3	$TiO_2 + 1$ wt% Fe_2O_3	$(8.0 \pm 0.4) \times 10^{-3}$	34.4 ± 1.7
	hydrate		
4	$TiO_2 + 1$ wt% Fe_2O_3	$(1.2 \pm 0.1) \times 10^{-2}$	68.2 ± 3.4
5	$TiO_2 + 0.97$ wt% AgCl	$(1.4 \pm 0.1) \times 10^{-2}$	73.5 ± 3.7
6	$TiO_2 + 0.78$ wt% Ag_2O	$(7.7 \pm 0.4) \times 10^{-3}$	65.0 ± 3.3

In contrast, the coatings of AgCl and Ag_2O cannot transfer electrons or holes to the acrylate in the photoexcited state and so, in these cases, a reduction in the photocatalytic activity of the sample would be expected (like in the case of Al_2O_3 coated TiO_2); for the sample coated with Ag_2O this was observed. However, in the case of the TiO_2 coated with AgCl, enhanced photocatalytic activity was recorded. This can be explained in terms of charge transfer between the TiO_2 and the AgCl. The photoexcited TiO_2 is able to transfer holes to the AgCl. From a chemical point of view this hole transfer corresponds to the oxidation of chloride ions to chlorine atoms; as chlorine atoms are reactive radical species they should be able to initiate the acrylate polymerization.

The results show that for an explanation of the photocatalytic activity of a coated sample the thermodynamics of the charge transfer between the basis material, the coating material and the reactants must be discussed. Moreover, charge transfer between the basis photocatalyst and the coating material must also be taken into account.

4. Conclusions

Coating TiO₂ with a small amount of inorganic compounds can enhance or reduce its ability to initiate acrylate

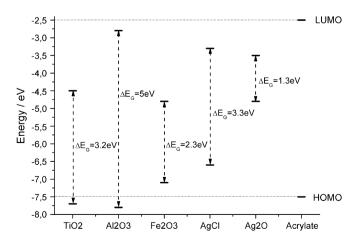


Fig. 6. Energy levels of the upper edge of the valence band (lower bar), the lower edge of the conduction band (upper bar) and band gap energies $\Delta E_{\rm G}$ for TiO₂ and the coating materials used in this work [16–21]. The locations of the highest occupied molecular orbital (HOMO) and lowest unoccupied molecular orbital (LUMO) of the acrylate are shown on the right side of the scheme.

polymerization. Thus, the coating of photocatalytic active pigments can be a way to achieve photopolymerization processes without recourse to a molecular photoinitiator. However, the polymerization rates achieved using coated TiO₂ pigments are low in comparison to samples containing photoinitiators. To substitute photoinitiators with photocatalytic active pigments, more research is needed to improve the photocatalytic activity of the pigments. One way to achieve this could be to optimise the amount of Fe₂O₃ or AgCl coating used as the affect of the coating material on the photocatalytic activity of the TiO₂ corresponding to that of its photoelectric properties, as proven by Photo-EMF measurements. This can be explained on the basis of the thermodynamics of charge transfer between the photoexcited catalyst samples and the acrylate.

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